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## INTERNATIONAL APPLICATION PUBLISHED UNDER THE PATENT COOPERATION TREATY (PCT)

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<b>(21) International Application Number:</b> PCT/CA93/00556 <b>(22) International Filing Date:</b> 17 December 1993 (17.12.93)  <b>(30) Priority Data:</b> 07/994,346           21 December 1992 (21.12.92)   US 08/167,188       16 December 1993 (16.12.93)   US  <b>(71) Applicant:</b> ONTARIO HYDRO [CA/CA]; 700 University Avenue, Toronto, Ontario M5G 1X6 (CA).  <b>(72) Inventor:</b> PALUMBO, Gino; 9 Tyler Place, Etobicoke, Ontario M9R 1L8 (CA).  <b>(74) Agents:</b> MITCHELL, Randall, S. et al.; Ridout & Maybee, 101 Richmond Street West, Suite 2300, Toronto, Ontario M5H 2J7 (CA).		<b>(81) Designated States:</b> CA, JP, KR, European patent (AT, BE, CH, DE, DK, ES, FR, GB, GR, IE, IT, LU, MC, NL, PT, SE).  <b>Published</b> <i>With international search report.</i> <i>Before the expiration of the time limit for amending the claims and to be republished in the event of the receipt of amendments.</i>
<b>(54) Title:</b> THERMOMECHANICAL PROCESSING OF METALLIC MATERIALS  <b>(57) Abstract</b>  <p>In the fabrication of components from a face centered cubic alloy, wherein the alloy is cold worked and annealed, the cold working is carried out in a number of separate steps, each step being followed by an annealing step. The resultant product has a grain size not exceeding 30 microns, a "special" grain boundary fraction not less than 60 %, and major crystallographic texture intensities all being less than twice that of random values. The product has a greatly enhanced resistance to intergranular degradation and stress corrosion cracking, and possesses highly isotropic bulk properties.</p>		

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**THERMOMECHANICAL PROCESSING OF METALLIC MATERIALS****CROSS-REFERENCE TO RELATED APPLICATION**

This application is a continuation-in-part of application No. 07/994,346 filed December 21, 1992 and entitled "Thermomechanical Processing of Metallic Materials".

**FIELD OF THE INVENTION**

This invention relates generally to the fabrication of alloy components wherein the alloy is subjected to cold working and annealing during the fabrication process. The invention is particularly addressed to the problem of intergranular degradation and fracture in articles formed of austenitic stainless alloys. Such articles include, for example, steam generator tubes of nuclear power plants.

**BACKGROUND OF THE INVENTION**

Intergranular degradation and fracture are among the commonest failure modes which currently compromise nuclear steam generator reliability. Previous attempts to alleviate susceptibility to intergranular failure have primarily involved the control of the alloy chemistry and the operating environment. However, the known source of the problem, the grain boundaries in the alloy, has largely been ignored.

The inventor and others have conducted studies to evaluate the viability of improving the resistance of conventional iron and nickel-based austenitic alloys, i.e. austenitic stainless alloys, to intergranular stress corrosion cracking (IGSCC) through the utilization of grain boundary design and control processing considerations. (See G. Palumbo, P.J. King, K.T. Aust, U. Erb and P.C. Lichtenberger, "Grain Boundary Design and Control for

Intergranular Stress Corrosion Resistance", Scripta Metallurgica et Materialia, 25, 1775 (1991)). The study produced a geometric model of crack propagation through active intergranular paths, and the model was used to evaluate the potential effects of "special" grain boundary fraction and average grain size on IGSCC susceptibility in equiaxed polycrystalline materials. The geometric model indicated that bulk IGSCC resistance can be achieved when a relatively small fraction of the grain boundaries are not susceptible to stress corrosion. Decreasing grain size is shown to increase resistance to IGSCC, but only under conditions in which non-susceptible grain boundaries are present in the distribution. The model, which is generally applicable to all bulk polycrystal properties which are dependent on the presence of active intergranular paths, showed the importance of grain boundary design and control, through material processing, and showed that resistance to IGSCC could be enhanced by moderately increasing the number of "special" grain boundaries in the grain boundary distribution of conventional polycrystalline alloys.

"Special" grain boundaries are described crystallographically by the well established CSL (coincidence site lattice) model of interface structure as those lying within  $\Delta \theta$  of  $\Sigma$ , where  $\Sigma \leq 29$ , and  $\Delta \theta \leq 15^\circ$  [see Kronberg and Wilson, Trans.Met.Soc. A.I.M.E., 1.85, 501 (1949) and Brandon, Acta Metall., 14, 1479 (1966)].

#### SUMMARY OF THE INVENTION

The present invention provides a mill processing methodology for increasing the "special" grain boundary fraction, and commensurately rendering face-centered cubic alloys highly resistant to intergranular degradation. The mill process described also yields a highly random distribution of crystallite orientations leading to

isotropic bulk properties (e.g., mechanical strength) in the final product. Comprehended within the term "face-centered cubic alloy" as used in this specification are those iron-, nickel- and copper-based alloys in which the principal metallurgical phase (>50% of volume) possesses a face-centered cubic crystalline structure at engineering application temperatures and pressures. This class of materials includes all chromium-bearing iron- or nickel-based austenitic alloys.

According to one aspect of the present invention, the method of enhancing the resistance of an austenitic stainless alloy to intergranular degradation comprises cold working the alloy to achieve a forming reduction less than the total forming reduction required, and usually well below the limits imposed by work hardening, annealing the partially reduced alloy at a temperature sufficient to effect recrystallization without excessive grain growth, and repeating the cold working and annealing steps cyclically until the total forming reduction required is achieved. The resultant product, in addition to an enhanced "special" grain boundary fraction and corresponding intergranular degradation resistance, also possesses an enhanced resistance to "sensitization". Sensitization refers to the process by which chromium carbides are precipitated at grain boundaries when an austenitic stainless alloy is subjected to temperatures in the range 500°C.-850°C. (e.g. during welding), resulting in depletion of the alloyed chromium and enhanced susceptibility to various forms of intergranular degradation.

By "cold working" is meant working at a temperature substantially below the recrystallization temperature of the alloy, at which the alloy will be subjected to plastic flow. This will generally be room temperature in the case of austenitic stainless alloys, but in certain circumstances

the cold working temperature may be substantially higher (i.e. warm working) to assist plastic flow of the alloy.

By "forming reduction" is meant the ratio of reduction in cross-sectional area of the workpiece to the original cross-sectional area, expressed as a percentage or fraction. It is preferred that the forming reduction applied during each working step be in the range 5%-30%, i.e..05-.30.

According to another aspect of the invention, in a fabricated article of formed face-centered cubic alloy having an enhanced resistance to intergranular degradation, the alloy has a grain size not exceeding 30 microns and a special grain boundary fraction not less than 60%.

#### BRIEF DESCRIPTION OF THE DRAWINGS

Preferred embodiments of the invention are described in detail below with reference to the drawings, in which:

Fig. 1 is a schematic representation of differences in texture components and in intensities determined by X-ray diffraction analysis between samples of UNS N06600 plate processed conventionally and by the process of the present invention;

Fig. 2 is a graphical comparison of the theoretically predicted and experimentally determined stress corrosion cracking performance of stressed UNS N06600 C-rings;

Fig. 3 is a graphical comparison between conventionally worked UNS N06600 plates and like components subjected to the process of the present invention, showing improved resistance to corrosion resulting from a greater percentage of special grain boundaries; and

Fig. 4 is an optical photomicrograph of a section of UNS N06600 plat produced according to the process of the invention.

#### PREFERRED EMBODIMENTS OF THE INVENTION

The method of the invention is especially applicable to the thermomechanical processing of austenitic stainless alloys, such as stainless steels and nickel-based alloys, including the alloys identified by the Unified Numbering System as N06600, N06690, N08800 and S30400. Such alloys comprise chromium-bearing, iron-based and nickel-based face-centered cubic alloys. The typical chemical composition of Alloy N06600, for example is shown in Table 1.

TABLE 1

<u>Element</u>	<u>% By Weight</u>
Al	ND
C	0.06
Cr	15.74
Cu	0.26
Fe	9.09
Mn	0.36
Mo	ND
Ni	74.31
P	ND
S	0.002
Si	0.18
Tl	ND

In the fabrication of nuclear steam generator tubing by thermomechanical processing according to the present invention a tubular blank of the appropriate alloy, for example Alloy N06600, is cold drawn and thereafter annealed. The conventional practice is to draw the tubing to the required shape in usually one step, and then anneal it, so as to minimize the number of processing steps. However, as is well known, the product is susceptible to intergranular

degradation. Intergranular degradation is herein defined as all grain boundary related processes which can compromise performance and structural integrity of the tubing, including intergranular corrosion, intergranular cracking, intergranular stress corrosion cracking, intergranular embrittlement and stress-assisted intergranular corrosion.

In contrast to current mill practice, which seeks to optimize the process by minimizing the number of processing steps, the method of the present invention seeks to apply a sufficient number of steps to yield an optimum microstructure. The principle of the method is based on the inventor's discovery that selective recrystallization induced at the most highly defective grain boundary sites in the microstructure of the alloy results in a high probability of continual replacement of high energy disordered grain boundaries with those having greater atomic order approaching that of the crystal lattice itself. The aim should be to limit the grain size to 30 microns or less and achieve a "special" grain boundary fraction of at least 60%, without imposing strong preferred crystallographic orientations in the material which could lead to anisotropy in other bulk material properties.

In the method of fabricating the tubing according to the present invention, the drawing of the tube is conducted in separate steps, each followed by an annealing step. In the present example the blank is first drawn to achieve a forming reduction which is between 5% and 30%, and then the partially formed product is annealed in a furnace at a temperature in the range 900-1050°C. The furnace residence time should be between 2 and 10 minutes. The temperature range is selected to ensure that recrystallization is effected without excessive grain growth, that is to say, so that the average grain size will not exceed 30µm. This average grain size would correspond to a minimum ASTM Grain



Size Number (G) of 7. The product is preferably annealed in an inert atmosphere, in this example argon, or otherwise in a reducing atmosphere.

After the annealing step the partially formed product is again cold drawn to achieve a further forming reduction between 5% and 30% and is again annealed as before. These steps are repeated until the required forming reduction is achieved.

There must be at least three cold drawing/annealing cycles to produce tubing having the required properties. Ideally the number of cycles should be between 3 and 7, there being little purpose in increasing the number of cycles beyond 7 since further cycles add but little to the fraction of resulting "special" grain boundaries. It will be noted that the amount of forming reduction per drawing step is given by

$$(1-r_t) = (1-r_i)^n$$

where  $r_i$  is the amount of forming reduction per step,  
 $r_t$  is the total forming reduction required,  
 $n$  is the number of steps, i.e. recrystallization steps.

The cold drawing of the tubing should be carried out at a temperature sufficient for inducing the required plastic flow. In the case of Alloy 600 and other alloys of this type, room temperature is usually sufficient. However, there is no reason why the temperature should not be well above room temperature.

A specific example of a room temperature draw schedule according to the invention as applied to UNS N06600 seamless tubing is given in the following Table 1. The total (i.e. cumulative) forming reduction which was required for the

article in this example was 68.5%. Processing according to the present invention involves annealing the tubing for three minutes at 1000°C between each forming step. This stands in contrast to the conventional process which applies the full 68.5% forming reduction prior to annealing for three minutes at 1000°C.

**Table 2**

STEP	OUTSIDE DIAMETER, mm	WALL THICKNESS mm	CROSS SECTIONAL AREA, mm <sup>2</sup>	% RA/step
Starting Dimensions	25.4	1.65	123.1	-
1	22.0	1.55	99.6	19.8
2	19.0	1.45	80.0	19.7
3	16.6	1.32	63.4	20.8
4	15.2	1.14	50.3	20.6
5	12.8	1.05	38.8	23.0

In Table 2 above, % RA/step refers to the percentage reduction in cross-sectional area for each of the five forming steps of the process. The cumulative forming reduction of  $r_t = 68.5\%$  is given by the aforementioned formula relating  $r_t$  to the amount of forming reduction per step,  $r_i$  and  $n$ , the total number of recrystallization steps.

In the resultant product, the alloy is found to have a minimized grain size, not exceeding 30 microns, and a "special" grain boundary fraction of at least 60%.

The above example refers particularly to the important application of fabricating nuclear steam generator tubing in

which the material of the end product has a grain size not exceeding 30 microns and a special grain boundary fraction of at least 60%, imparting desirable resistance to intergranular degradation. However, the method described is generally applicable to the enhancement of resistance to intergranular degradation in Fe - Ni - and Cu -based face-centered cubic alloys which are subjected to forming and annealing in fabricating processes.

Thus, in the fabrication of other Fe-, Ni-, and Cu- based face-centered cubic alloy products by rolling, drawing, or otherwise forming, wherein a blank is rolled, drawn or formed to the required forming reduction and then annealed, the microstructure of the alloy can be greatly improved to ensure the structural integrity of the product by employing a sequence of cold forming and annealing cycles in the manner described above.

In Table 3 below, two examples, tubing and plate, are given for comparing the grain boundary distributions in alloy UNS N06600 arising from "conventional process" (that is, one or two intermediate annealing steps) and the present "New Process" which involves multiple processing steps ( $\geq 3$ ):

**Table 3**

Material:	UNS N06600 Tubing - Conventional Process	UNS N06600 Tubing - New Process	UNS N06600 Plate - Conventional Process	UNS N06600 Plate - New Process
Total No:	105	96	111	102
I1	1	0	4	2
I3	34	48	26	47
I5	2	1	0	0
I7	1	1	0	1
I9	2	13	7	10
I11	1	1	0	2
I13	0	1	2	0
I15	3	1	0	0
I17	1	0	0	0
I19	1	0	1	0
I21	1	1	0	2
I23	0	0	0	0
I25	1	0	1	1
I27	3	7	0	7
I29	0	0	0	0
I>29 (General)	64	22	70	30
% Special (I≤29)	48.6%	77.1%	36.9%	70.6%

To afford a basis for comparison, the total forming reduction for tube processing (columns 2 and 3 of Table 3) and plate processing (columns 4 and 5 of Table 3) is again 68.5% in each case. In the conventional process, that degree of total forming reduction has been achieved in one single step with a final anneal at 1000°C for three minutes and, in the new process, in five sequential steps involving 20% forming reduction per step, with each step followed by annealing for three minutes at 1000°C. The numerical entries are grain boundary character distributions  $\Sigma 1$ ,  $\Sigma 3$  etc. determined by Kikuchi diffraction pattern analysis in a scanning electron microscope, as discussed in v. Randle, "Microtexture Determination and its applications", Inst. of Materials, 1992 (Great Britain). The special grain boundary fraction for the conventionally processed materials is 48.6% for tubing and 36.9% for plate, by way of contrast with respective values of 77.1% and 70.6% for materials treated by the new forming process.

As illustrated in Figure 1, the randomization of texture by processing according to the present invention leads to wrought products having highly uniform bulk properties. Figure 1 shows in bar graph form the differences in texture components and intensities determined by X-ray diffraction analysis between UNS N06600 plate processed conventionally (single 68.5% forming reduction followed by a single 3 minute annealing step at 1000°C) and like material treated according to the new process (68.5% cumulative forming reduction using 5 reduction steps of 20% intermediate annealing for 3 minutes at 1000°C).

The major texture components typically observed in face-centered cubic materials are virtually all eliminated with the new process; the exception being the Goss texture  $[110]\langle 001 \rangle$  which persists at just above that expected in a random distribution (i.e., texture intensity of 1). The new process

thus yields materials having a highly desirable isotropic character.

As illustrated in Figure 2, wrought products subjected to the process of the present invention possess an extremely high resistance to intergranular stress corrosion cracking relative to their conventionally processed counterparts. The graph of Figure 2 summarizes theoretical and experimental stress corrosion cracking performance as it is affected by the population of "special" grain boundaries in the material. The experimental results are for UNS N06600 C-rings stressed to 0.4% maximum strain and exposed to a 10% sodium hydroxide solution at 350°C for 3000 hours. The dashed line denotes the minimum special grain boundary fraction of 60% for fabricated articles according to the present invention.

In addition to displaying a significantly enhanced resistance to intergranular corrosion in the as-processed mill annealed condition, wrought stainless alloys according to the present invention also possess a very high resistance to sensitization. This resistance to carbide precipitation and consequent chromium depletion, which arises from the intrinsic character of the large population of special grain boundaries, greatly simplifies welding and post-weld procedures and renders the alloys well-suited for service applications in which temperatures in the range of 500°C to 850°C may be experienced. Figure 3 summarizes the effect of special grain boundary fraction on the intergranular corrosion resistance of UNS N06600 plates as assessed by 72-hour testing in accordance with ASTM G28 ("Detecting Susceptibility To Intergranular Attack in Wrought Nickel-Rich, Chromium Bearing Alloys").

As shown in Figure 3, materials produced using the new process (in which the special grain boundary fraction exceeds 60%) display significantly reduced corrosion rates over those

produced using conventional processing methods. Furthermore, the application of a sensitization heat treatment (i.e. 600°C for two hours) to render the materials more susceptible to intergranular corrosion by inducing the precipitation of grain boundary chromium carbides, has a far lesser detrimental affect on materials having high special boundary fractions, i.e. those produced according to the process of the present invention.

The high special boundary fraction exhibited in a UNS N06600 plate which has been produced using the process of the invention may be directly visually appreciated from Figure 4, an optical photomicrograph of a Section of such plate (210X magnification). The good "fit" of component crystallite boundaries is evident by the high frequency of annealing twins, which appear as straight boundary lengths intersecting other boundaries at right angles.

It should be finally pointed out that, although the method of the present invention differs from conventional mill practice which seeks to minimize the number of forming and annealing steps, it is otherwise perfectly compatible with existing mill practice in that it does not call for changes in the equipment used.

## I CLAIM:

1. In the fabrication of articles from a face-centered cubic Fe-, Ni-, or Cu- based alloy wherein the alloy is subjected to cold working and annealing steps to effect a desired total forming reduction, a method of enhancing resistance of the alloy to intergranular degradation, which comprises

cold working the alloy to achieve a forming reduction less than the total forming reduction desired,

annealing the partially reduced alloy at a temperature sufficient to effect recrystallization while limiting grain size to 30 microns or less, and

repeating said cold working and annealing steps cyclically until the total forming reduction required is achieved.

2. A method according to claim 1, in which the alloy is an austenitic stainless alloy selected from the group consisting of chromium bearing iron-based and nickel-based face-centered cubic alloys.

3. A method according to claim 2, in which the alloy is Alloy N06600 (Ni-16Cr-9Fe).

4. A method according to any of the previous claims, in which the applied forming reduction per cycle is in the range 5%-30%.

5. A method according to any of the previous claims, in which each annealing step is conducted in a furnace at a temperature in the range 900-1050°C. for a time 2-10

minutes.

6. A method according to any of the previous claims, in which the annealing is conducted in an inert or a reducing atmosphere.

7. In the fabrication of articles from an austenitic stainless, iron-based or nickel-based face-centered cubic alloy wherein the alloy is subjected to cold working and annealing steps, the improvement which comprises subjecting the alloy to successive cold working and annealing cycles whereby to enhance the resistance of the alloy to intergranular degradation, each said cycle consisting of

- i) a cold working step in which the alloy is subjected to a forming reduction between 5% and 30%, and
- ii) an annealing step in which the reduced alloy is annealed at a temperature in the range 900-1050°C. for a time 2-10 minutes, said cycles being repeated until a desired total forming reduction is achieved.

8. A method according to claim 7, in which each cold working step is a cold drawing step.

9. A method according to claim 7, in which each cold working step is a cold rolling step.

10. A method according to claim 7, claim 8 or claim 9, in which the annealing steps are conducted in an inert or a reducing atmosphere.



11. In the fabrication of steam generator tubing from a blank of austenitic stainless alloy, the method of enhancing the intergranular cracking resistance of the alloy which comprises subjecting the blank to successive cold drawing and annealing cycles, each said cycle consisting of

- i) a cold drawing step in which the blank is subjected to a forming reduction between 5% and 30%, and
- ii) an annealing step in which the blank is annealed in an inert or a reducing atmosphere at a temperature in the range 900-1050°C. for a time 2-10 minutes, said cycles being repeated until a desired total forming reduction is achieved.

12. A method according to claim 11, in which the alloy is selected from the group consisting of N06600, N06690, N08800 and S30400.

13. An article fabricated according to the method of any one of claims 1 to 7, wherein said face-centered cubic alloy has a "special" grain boundary fraction not less than 60%.

14. An article according to claim 13, wherein said alloy exhibits crystallographic texture intensities less than twice the values expected in a random distribution.

15. An article according to claim 14, wherein the grain size of said alloy does not exceed 30 microns.

16. A fabricated article according to claim 15, wherein said article is a section of steam generator tubing of formed austenitic stainless alloy.

17. A fabricated article according to claim 15, wherein said austenitic stainless alloy is Alloy N06600 (Ni-16Cr-9Fe).

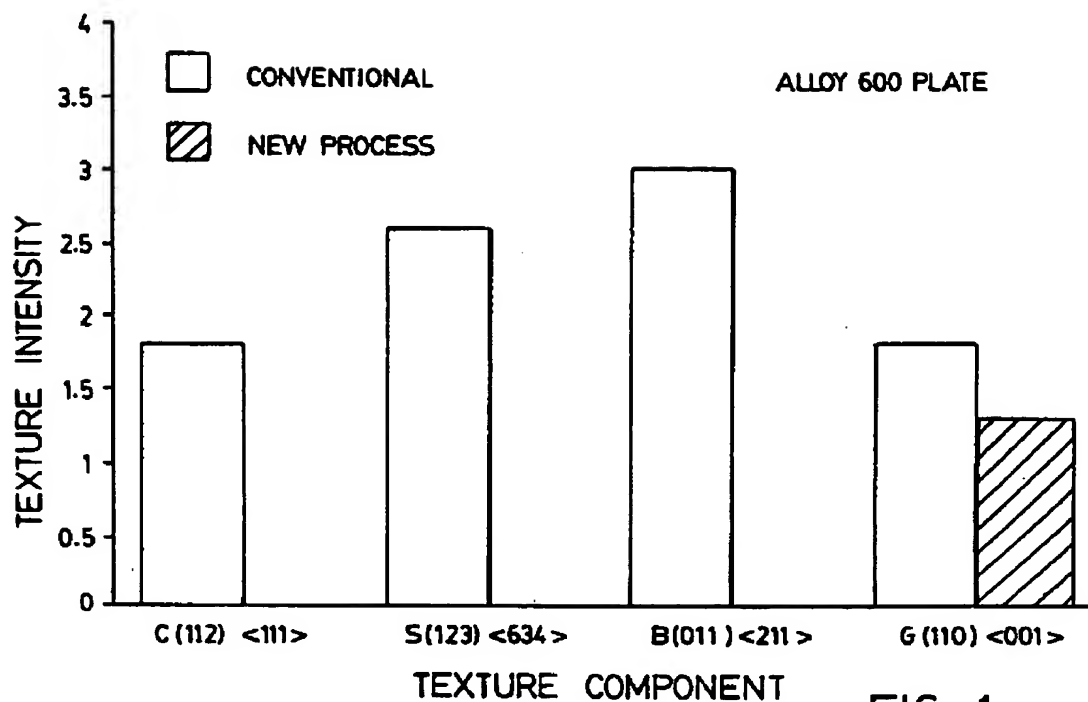


FIG. 1

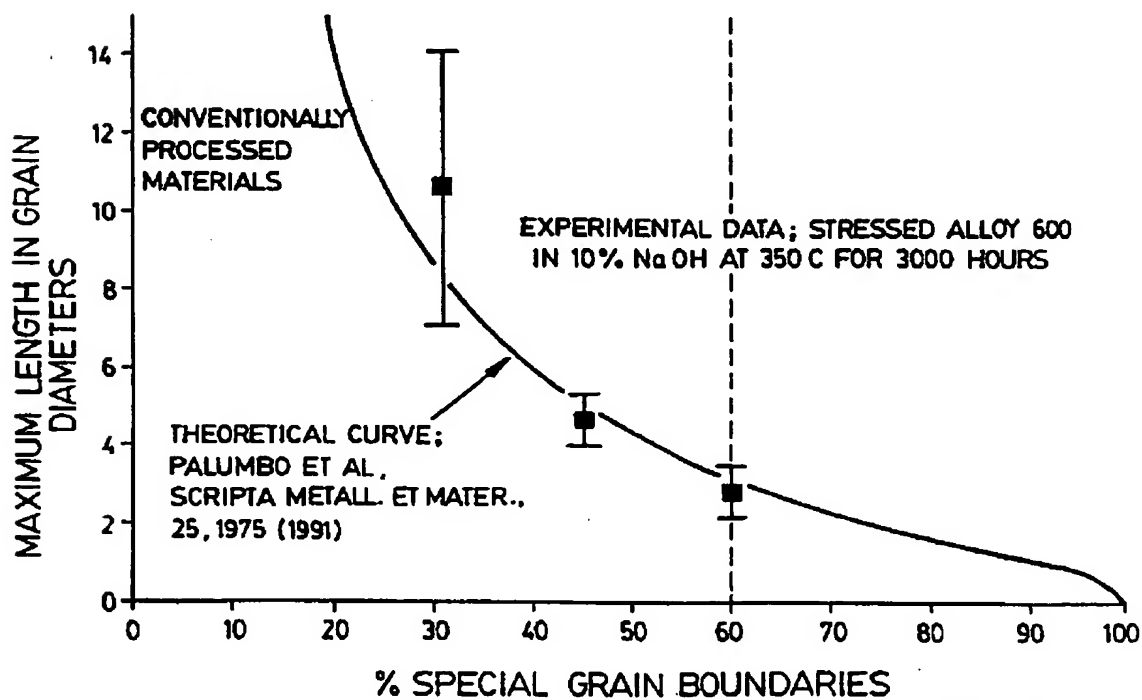
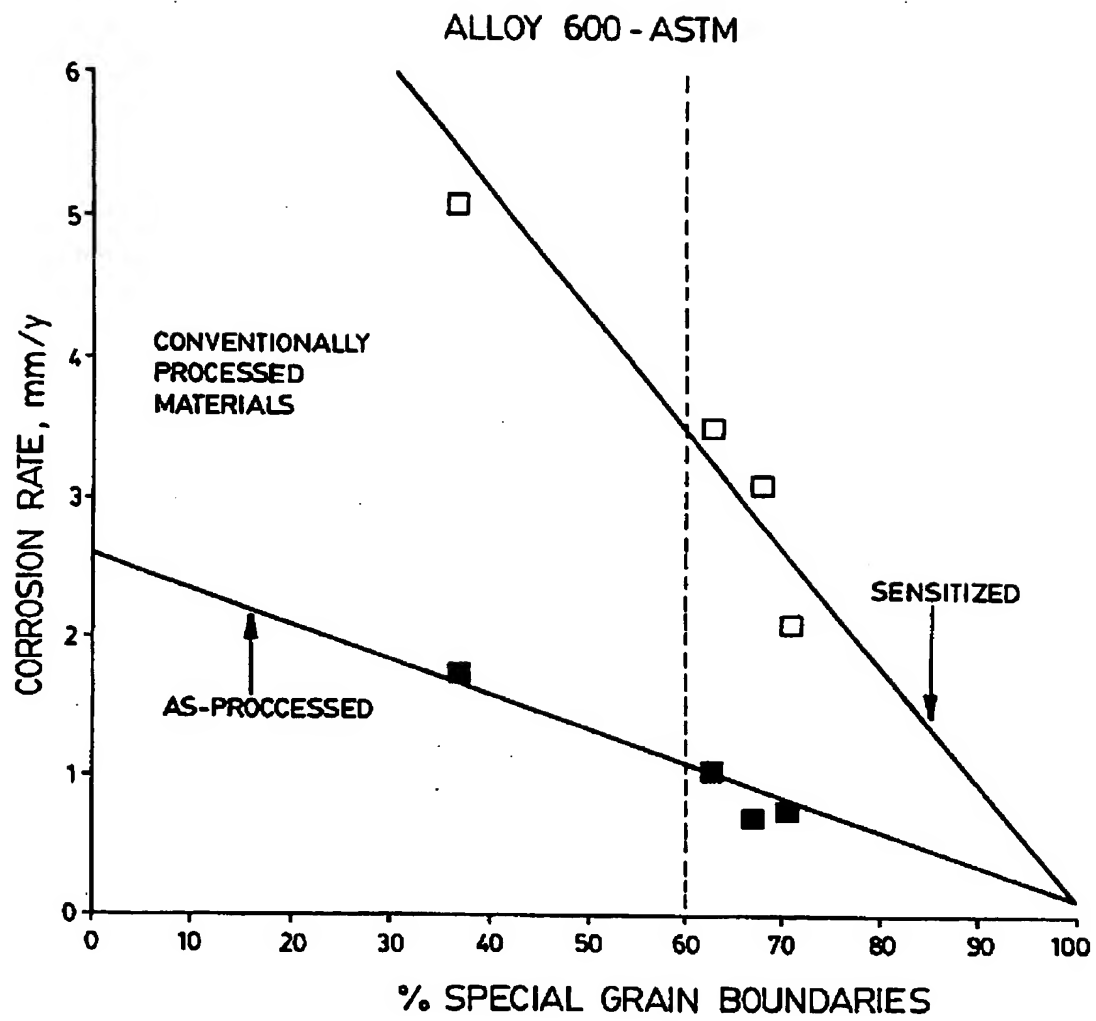


FIG. 2

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FIG. 3

2/3

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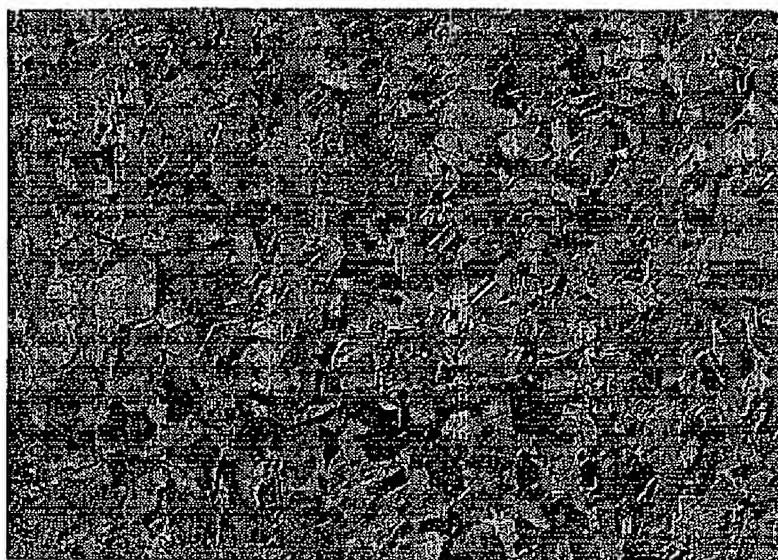


FIG. 4

## INTERNATIONAL SEARCH REPORT

International Application No  
PCT/CA 93/00556

## A. CLASSIFICATION OF SUBJECT MATTER

IPC 5 C21D8/00 C21D8/10 C22F1/10 C22F1/08

According to International Patent Classification (IPC) or to both national classification and IPC

## B. FIELDS SEARCHED

Minimum documentation searched (classification system followed by classification symbols)

IPC 5 C22F C21D

Documentation searched other than minimum documentation to the extent that such documents are included in the fields searched

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## C. DOCUMENTS CONSIDERED TO BE RELEVANT

Category *	Citation of document, with indication, where appropriate, of the relevant passages	Relevant to claim No.
X	GB,A,1 124 287 (UNITED KINGDOM ATOMIC ENERGY AUTHORITY) 21 August 1968	1,2
A	see page 1, line 11 - line 14; claim 1; table 1	7,11
A	GB,A,2 027 627 (KERNFORSCHUNGSZENTRUM KARLSRUHE GMBH) 27 February 1980	1,7,8,11
A	US,A,4 070 209 (M.USUI) 24 January 1978	1,7,11
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☒ Further documents are listed in the continuation of box C.☒ Patent family members are listed in annex.

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Date of the actual completion of the international search

12 April 1994

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# INTERNATIONAL SEARCH REPORT

Int'l Application No  
PCT/CA 93/00556

## C.(Continuation) DOCUMENTS CONSIDERED TO BE RELEVANT

Category	Citation of document, with indication, where appropriate, of the relevant passages	Relevant to claim No.
A	US,A,3 855 012 (S. SHAPIRO ET AL) 17 December 1974 see claims 1-3 ---	1
A	FR,A,1 475 970 (UNITED KINGDOM ATOMIC ENERGY AUTHORITY) 7 April 1967 see example 3 ---	1,7,11
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A	PATENT ABSTRACTS OF JAPAN vol. 11, no. 229 (C-436)25 July 1987 & JP,A,62 040 336 (MITSUBISHI METAL CORP) 21 February 1987 see abstract ---	
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Information on patent family members

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